Correlation Between Permeability, Losses and Compact Formation in a NiZn Ferrite

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The correlation between initial permeability μ_i , loss factor $tan\delta$ and powder compaction piessure in a ferrite of nominal coniposition Ni_{0.33}Zn_{0.64}Fe_{2.06}O_{4+ α} synthesized by ceramic methods is shown, for two groups of samples with different amounts of binder added. The least pressure necessary to obtain the best magnetic properties in both cases was about **3** ton/cm². However, this behavior is not reflected ill any singularity of the green density curves as should be expected when compasing with other inaterials, not allowing the use of these curves alone as a measure of coinpact quality before firing in NiZn ferrites.

I. Introduction

The microstructure parameters of polycrystalline magnetic materials usually have a decisive influence on magnetic properties. A given chemical composition may originate very different properties in dependence of the conditions at the grain boundaries, the grain size and pore distilbutions and the presence or not of microinhomogeneities. The microstructure is controlled by a great number of parameters of the synthesis procedure, such as the thermal treatments, compact formation and, when solid reaction synthesis is carried out, morphology ar d grain distribution of the powdered raw rnaterials and milling^[1-4].

In this case of ferrites obtained by sintering the previously reacted and compacted powdered material, the parameters related to the compactation process have a strong influence on microstructure and on magnetic properties. Examples are the differences obtained when the compactat on is carried out by isostatic pressing or by hot-pressing; instead of the conventional die-pressing. Die-pressing is the most important one for practical purposes, due to its relative simplicity^[2].

One of the parameters to take into account when the compact is formed is the so called "green" density. In many inaterials it usually increases almost linearly with the compaction pressure to a given value, where a change of slope takes place. This is attributed to fractures or plastic deformation in the grains which appears when the elastic limit of the powdered material is exceeded. In the case of ceramic materials, another parameter which must be taken into account is the amount of binder added, essential to obtain a green compact stiff enougli so as to keep its shape in further manipulations^[5,6]. The aim of the present research was to study tlie correlation between the green density and tlie best magnetic properties of tlie NiZn ferrite samples.

II. Experimental

Samples of NiZn ferrite witli composition $Ni_{0.33}Zn_{0.64}Fe_{2.06}O_{4+\alpha}$ were prepared using conventional ceramic techniques from analytical reagents, heating the powders at 900 °C after mixing wet in a 50% ethanol aqueous solution for 1 hour in a vertical steel ball mill. The calcined powders were wet milled again for tlie same time, and compacts with good stiffeness were obtained by die-pressing in the form of square section toroids, witli external and internal diameter of 23 and 11 mm respectively, adding polyvinyl alcohol dissolved in water as binder to a extent of 1.5 dry wt% (A samples) and 2% (B samples). Before coinpacting, the mixed powders were forced through a sieve with 1 mm² openings. The toroids were sintered at a maximum temperature of 1200 °C (1 hour) witli a rate of heating and cooling of $300 \,^{\circ}$ C/h from room temperature^[7]. The green and fired densities of tlie compacts were measured in air, as well as the initial relative permeability μ_i and tlie loss factor $tan\delta$ of tlie fired samples, using a Tesla BM-560 Q-meter.

III. Results and Discussion

The variation of μ_i as a function of the applied compacting pressure appears in Fig. 1. In both cases the permeability increased almost linearly with the applied pressure up to about the same value of 3.5 toii/cm^2 , where it tended to stabilize. This behaviour is reflected in the fired density of the two groups of samples, where a slight stabilization or change of slope is reached at about 3-4 ton/cm² (see Fig. 2). Losses also sliow a minimum in the same range of pressures; liowever, the surpassing of 3.5 ton/cm^2 was deleterious for the B samples (Fig. 3). In all figures the drawn curves are the result of adjustiling the experimental points to 2nd and 3rd degree polynomials by numerical methods.



Figure 1: Relative permeability μ_i vs applied compaction pressure; A) 1.5 wt%; B) 2 wt% polyvinyl.



Figure 2: Green and fired densities in $Ni_{0.33}Zn_{0.64}Fe_{2.06}O_{4+\alpha}$ samples as a function of the applied compaction pressure. A) 1.5 wt%; B) 2 wt% polyvinyl.



Figure 3: $\tan \delta / \mu$ for the samples of Figure 2.

The difference in permeability for A and B samples in Fig. 1 suggests an appreciable dependence of μ_i on the amount of added polyvinyl in the range analyzed. This difference is also reflected in the green density curves of Fig. 2; however, the non-controlled humidity of the mixed powders and the fact that the difference in permeability is not reflected in the fired density curves did not allow us to attribute the observed beliavior to the differences in the humidity of the samples; i.e., differences in the samples in the same effect.

The least pressure necessary to obtain the hest magnetic properties in both cases was about 3 ton/cm^2 . However, this value was not reflected in aily singularity of the green density curves, not allowing the use of these curves alone as a measure of coinpact quality before firing.

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